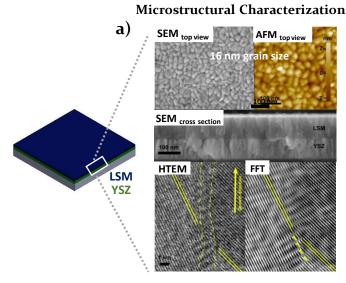
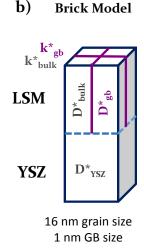
## Evaluation of the Mass Transport Properties of La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3+δ</sub> nanostructures by Finite Element Method (FEM) Simulations

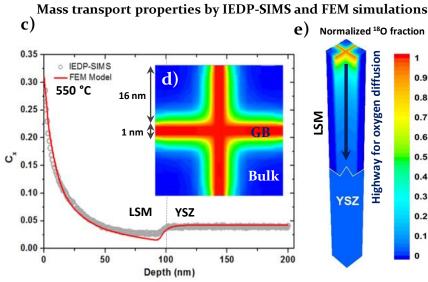
D. Pla<sup>1,2</sup>, A. Morata<sup>1</sup>, A. M. Saranya<sup>1</sup>, A. Cavallaro<sup>3</sup>, J. Canales-Vázquez<sup>4</sup>, F. Chiabrera<sup>1</sup>, J. A. Kilner<sup>3</sup>, M. Burriel<sup>2</sup>, and A. Tarancón<sup>1</sup> <sup>1</sup>IREC, Catalonia Institute for Energy Research, Spain. <sup>2</sup> Université Grenoble Alpes, CNRS, LMGP, France. <sup>3</sup> Imperial College London, London, UK. <sup>4</sup> Universidad de Castilla La Mancha, Spain

NANOENGINEERING THIN FILMS: nanostructures with a high density of grain boundaries (GBs)

 $La_{0.8}Sr_{0.2}MnO_{3\pm\delta}$  (LSM) dense film deposited by PLD on polycrystalline YSZ films







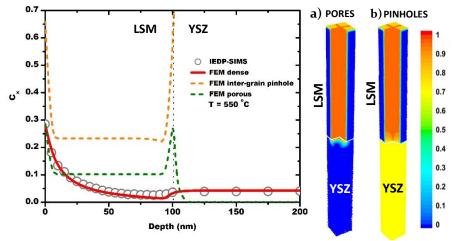
(a) The high density of vertically aligned GBs leads to a high concentration of strain-induced defects.

(b) Model proposed for the FEM simulations.

(c) Normalized <sup>18</sup>O isotopic fraction and fitted solution by FEM at 550°C. (d) Top view and (e) cross section of the isotopic fraction map by FEM.

## **EFFECT OF DEFECTS in diffusion profiles**

Short circuit paths down nanopores or inter-grain pinholes



Differences in the isotope fraction profiles at 550°C due to short circuit paths: (a) porosity in the place of GB and (b) 1x1nm<sup>2</sup> squared inter-grain pinholes inside GB. The presence of defects in the LSM dense thin films is discarded

**GEOMETRIC EFFECTS** 

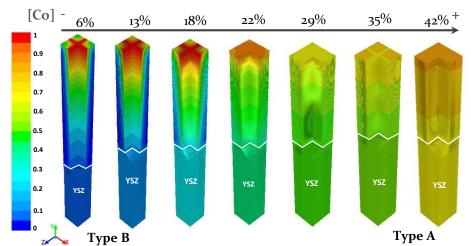
Variation of the grain size and the grain boundary width (according to the grain size study from the structural characterization)

	6% [C	o] LSMC	
	D* <sub>b</sub> (cm <sup>2</sup> ·s <sup>-1</sup> )	D* <sub>GB</sub> (cm <sup>2</sup> ·s <sup>-1</sup> )	k* <sub>GB</sub> (cm·s <sup>-1</sup> )
Grain wid	th (nm) and 1 r	ım GB	
15	3.5x10 <sup>-17</sup>	3.9x10 <sup>-12</sup>	2.6x10 <sup>-4</sup>
21	7.0x10 <sup>-17</sup>	3.5x10 <sup>-12</sup>	2.0x10 <sup>-4</sup>
27	1.1x10 <sup>-16</sup>	3.2x10 <sup>-12</sup>	1.3x10 <sup>-4</sup>
GB width	(nm) and 21 nr	n grain	
0.5	9.8x10 <sup>-17</sup>	4.3x10 <sup>-12</sup>	3.0x10 <sup>-4</sup>
1.5	5.8x10 <sup>-17</sup>	2.1x10 <sup>-12</sup>	1.9x10 <sup>-4</sup>
	13% [0	Co] LSMC	
Grain wid	th (nm) and 1 r	ım GB	
18	5.5x10 <sup>-17</sup>	2.9x10 <sup>-12</sup>	8.2x10 <sup>-5</sup>
25	9.8x10 <sup>-17</sup>	2.8x10 <sup>-12</sup>	8.0x10 <sup>-5</sup>
32	1.7x10 <sup>-16</sup>	2.6x10 <sup>-12</sup>	7.9x10 <sup>-5</sup>
GB width	(nm) and 25 nr	n grain	
0.5	1.9x10 <sup>-16</sup>	2.7x10 <sup>-12</sup>	1.0x10 <sup>-4</sup>
1.5	8.0x10 <sup>-17</sup>	2.1x10 <sup>-12</sup>	7.9x10 <sup>-5</sup>

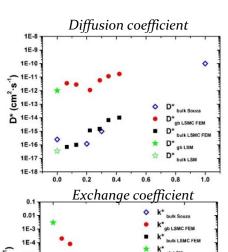
- ✓ The variation of the grain width mainly affects  $D_b^*$  and  $D_{GB}^*$ .
- The k\*<sub>GB</sub> value is much less sensitive.
- Changes in the k\*<sub>b</sub> value are insignificants.
- The effect of the grain boundary width modifies in the same order the  $k^*_{GB}$ ,  $D^*_{GB}$  and D\*<sub>h</sub> due to the decisive influence of GBs and their interaction with the bulk.

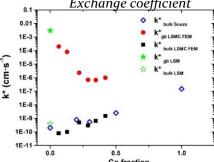
## DOPANT EFFECT : $La_{o.8}Sr_{o.2}Mn_{1-x}Co_xO_{3\pm\delta}(LSMC)$

Isotopic fraction maps for LSMC/YSZ calculated by FEM



An increase in Co concentration leads to a change of the diffusion regime according to the Harrison's classification from the type B ( $D^*_{b}$  and  $D^*_{GB}$ ) to the type A





- Large increase in both D\*, and  $D_{GR}^*$  for %Co > 20%, probably related to an increase of the Vo
- The increase of  $V_{0}^{\cdots}$  is similar in the grain and GBs.
- K\*, increases with Co content, probably related to an increase of the V<sub>O</sub><sup>..</sup> in the grain.
- $k^*_{GB}$  decreases with %Co up to 22%. It might be due to a higher E<sub>a</sub> for O<sub>2</sub> adsorption in LSMC compared to LSM.
- The O<sup>2-</sup> incorporation probably is enhanced due to high density of GBs in the nanostructure.

